

Membraneless Unbuffered Seawater Electrolysis for Pure Hydrogen Production

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Article

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Membraneless Unbuffered Seawater Electrolysis for Pure Hydrogen

Production

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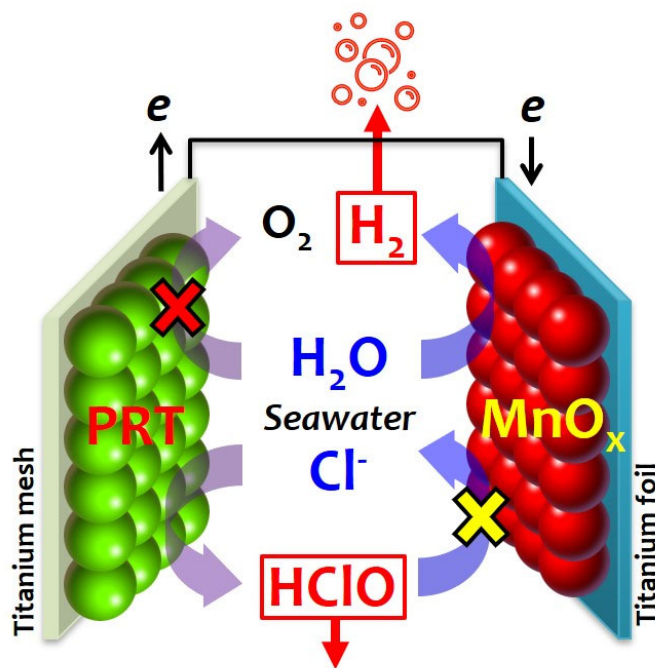
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Broader Context

Electrolytic hydrogen production is considered a key energy transition, carbon-neutral technology when coupled with renewable energy production processes. As the H₂ price is highly dependent on geographic location, the maritime transport of as-produced H₂ at low costs to locations where the H₂ production is unfavorable is an alternative. In this regard, seawater electrolysis is promising for *in situ* H₂ production and overseas transport. Although the conventional electrolysis has been extensively studied in well-defined aqueous environment characteristics (e.g., electrolyte and pH), such conditions cannot be directly applied to the seawater electrolysis at a pH ~8.2 because of the presence of several types of natural inorganic species. Conventional seawater electrolysis proceeds under alkaline conditions (pH > 13) to ensure the selectivity of the oxygen evolution reaction (OER) over the chloride oxidation reaction (ClOR). In addition, membranes are required to inhibit the chlorine reduction reaction (ClRR) at the cathode and maximize the hydrogen evolution reaction (HER) while separating H₂ from O₂. Herein, we propose a novel membraneless electrolysis process for the production of high-purity H₂ from unperturbed, unbuffered seawater at pH 8.2. To achieve this goal, we attempted to synthesize ClOR-selective anodes and ClRR-inhibitive cathodes, while completely retarding the OER at the anode and maximizing the HER at the cathode. The as-synthesized anode and cathode pair drove ClOR and HER with Faradaic efficiencies of ~100% over 100 h in a single-compartment cell containing seawater at pH 8.2. This technology is expected to enable *in situ* seawater electrolysis to produce high-purity H₂, which can be quickly collected and transported overseas.

Abstract

Seawater electrolysis is promising as an *in situ* H₂ production method, allowing immediate overseas transport of the produced H₂. Typically, it proceeds under alkaline conditions to ensure the selectivity of the oxygen evolution reaction (OER) over the chloride oxidation reaction (CIOR). In addition, electrolyzers usually require membranes to inhibit the chlorine reduction reaction (CIRR) and maximize the hydrogen evolution reaction (HER) while separating H₂ from O₂. In this study, membraneless electrolysis of undisturbed, unbuffered seawater (pH 8.2) is proposed to produce high-purity H₂. A ternary component catalyst with Pt, Ru, and Ti (PRT) and a minimized Pt level (Pt_{0.06}Ru_{0.24}Ti_{0.7}O_x) drives CIOR at a Faradaic efficiency (FE) of ~100% in saline water at an industrial level of current density ($J = 800 \text{ mA cm}^{-2}$) over 500 h. CIRR is also inhibited on MnO_x electrodes, leading to ~100% FE of H₂ production even in the presence of chlorines (HClO/ClO⁻). Finally, the PRT anode/MnO_x cathode pair drive CIOR and HER at FEs of ~100% at $J = 80 \text{ mA cm}^{-2}$ over 100 h in a single-compartment cell containing seawater. Moreover, the OER and CIRR are completely inhibited during seawater electrolysis.

Introduction

Electrolytic hydrogen production is considered a key energy transition, carbon-neutral technology when coupled with renewable energy production processes.¹⁻⁹ The minimum H₂ production cost was estimated to decrease from \$4.56 kg⁻¹ in 2020 to \$2.44 kg⁻¹ in 2050 upon direct connection with solar- and wind-powered grids.^{10,11} As the H₂ price is highly dependent on geographic location, the maritime transport of as-produced H₂ at low costs to locations where the H₂ production is unfavorable is an alternative. In this regard, seawater electrolysis is promising for *in situ* H₂ production and overseas transport. Harvesting renewable energy and directly connecting it to the electrolyzer further increases the technical readiness of seawater electrolysis.^{4,12,13}

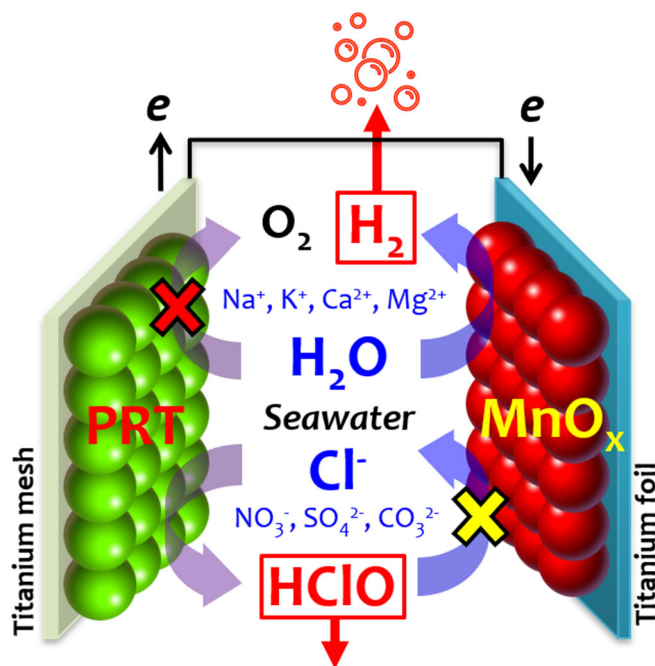
Although the conventional electrolysis has been extensively studied in well-defined aqueous environment characteristics (e.g., electrolyte and pH), such conditions cannot be directly applied to the seawater electrolysis at a pH ~8.2 because of the presence of several types of natural inorganic species.^{6,14-17} Among them, chlorides seriously interfere with oxygen and hydrogen evolution reactions (OER and HER, respectively).^{7,18,19} The electrochemical potentials of chloride oxidation reaction (ClOR, reaction 1 at $E^\circ = 1.482$ V) and OER ($E^\circ = 1.229$ V) are similar, but the ClOR with two-electron transfer becomes dominant,²⁰⁻²³ particularly at high current densities. In addition, the chlorine reduction reaction (ClRR, a reverse of ClOR) with the as-produced chlorines (represented by the HClO/ClO⁻ pair, $pK_a = 7.46$) competes with HER.^{8,9,19,22-24} The active involvement of ClOR/ClRR eventually reduces the Faradaic efficiencies (FEs) of O₂ and H₂ production.



To address this challenge, a membraned alkaline electrolysis system has often been employed for seawater electrolysis.^{14,25,26} An increase in the pH of seawater reduces both E°_{OER} and E°_{ClOR} to different degrees (e.g., $E^{\circ}_{\text{OER}} = 0.401$ V and $E^{\circ}_{\text{ClOR}} = 0.81$ at pH 14), and the OER becomes dominant over the ClOR.^{26,27} Even if ClOR occurs concurrently, the employed membranes separate evolved O₂ and H₂ gases as well as inhibit the crossover of HClO to the catholyte. Accordingly, the negative effects of ClOR/CIRR are minimized. Although feasible, however, the synthesis of OER-selective catalyst durably working at a large J in seawater is difficult. In addition, the use of membranes and corrosive KOH significantly increases the H₂ production cost.^{28,29} Furthermore, dissolved organic substances (~90% of the total organic carbon in the oceans) interfere with the OER and cause fouling of the membranes.^{30,31}

With this in mind, we propose a membraneless electrolysis of unperturbed, unbuffered seawater at pH 8.2 for the production of high-purity H₂. To achieve this goal, we attempted to synthesize ClOR-selective anodes and CIRR-inhibitive cathodes, while completely retarding the OER at the anode and maximizing the HER at the cathode (**Scheme 1**). If the ClOR and HER are coupled, high-purity H₂ gas without O₂ can be obtained with no need of a membrane because the as-formed chlorines remain in the aqueous phase. For selective ClOR and HER, ternary Pt, Ru, and Ti oxide (PRT) anodes and MnO_x cathodes were synthesized, respectively. An optimized PRT electrocatalyst with a minimized Pt level (Pt_{0.06}Ru_{0.24}Ti_{0.7}O_x) drove ClOR at an FE of ~100% in saline water at an industrial level of J (0.8 A cm⁻²) over 500 h. Furthermore, CIRR was effectively inhibited by the MnO_x cathode, leading to ~100% FE in the H₂ production. Finally, the PRT anode and MnO_x cathode pair produced aqueous free chlorine and H₂ gas with FEs of ~100% over 100 h in a single-compartment cell containing

seawater at pH 8.2.



Scheme 1. Illustration of seawater electrolysis with a Pt, Ru, and Ti (PRT) anode and MnO_x cathode pair in a membraneless single-compartment cell. Conventional seawater electrolysis proceeds under alkaline conditions ($\text{pH} > 13$) to secure selectivity of oxygen evolution reaction (OER) over chloride oxidation reaction (ClOR). In addition, it usually requires membranes to inhibit chlorine reduction reaction (ClRR) and maximize hydrogen evolution reaction (HER), while separating H_2 from O_2 . However, if the electrolysis products are only HClO (and/or ClO^-) in solution and H_2 in headspace at Faradaic efficiencies of $\sim 100\%$ for ClOR and HER, respectively, membranes for separating gases (H_2 from O_2) and inhibiting chlorine reduction at the cathode are not necessary.

Experimental

Synthesis and surface characterization of materials

Analytical grade chemical reagents and solvents were purchased from Sigma-Aldrich and used without further purification, unless otherwise mentioned. Titanium meshes (1 cm × 3 cm, 1-mm thick, Alfa Aesar) and foils (0.127-mm thick, 99.7% pure, Aldrich) were rubbed with sand paper, washed with deionized water (18 MΩ cm) and ethanol, and subsequently treated with aqueous HCl (37%) solution for 20 min at 50 °C. In sequence, the substrates were washed with excess deionized water and ethanol. For the synthesis of ternary (Pt_{0.2}Ru_{0.8})_xTi_{1-x}O_y (PRT-*x*) particles (*x* = 0.1–0.4), various ratios of H₂PtCl₆·H₂O (dissolved in 0.1 mL HCl), RuCl₃·H₂O (dissolved in 0.1 mL HCl), and Ti(C₄H₉O)₄ (dissolved in 0.5 mL ethanol) were mixed in a mixed solution of citric acid (C₆H₈O₇, Duksan, 99.5%) and ethylene glycol (C₂H₆O₂, Daejung, 99.5%) (1/4 by molar ratio) at 90 °C. After mixing for 5 min, the PRT samples were annealed at various temperatures (400–900°C) for 1 h in air. The as-obtained PRT powder (10 mg) was dispersed in 2-propanol (2 mL, C₃H₈O, Duksan, 99.9%) using ultrasound for 2 h. This PRT slurry was sprayed onto Ti meshes with an air gun at loading masses of 0.83–3.33 mg cm⁻². Subsequently, the PRT/Ti samples were dried overnight in air. Unless otherwise specified, PRT represents the sample for *x* = 0.3 (i.e., Pt_{0.06}Ru_{0.24}Ti_{0.7}O_y) annealed at 500 °C with a loading mass of 3.33 mg cm⁻². For comparison, binary Pt_{0.2}Ru_{0.8}O_y (PR, *i.e.*, PRT with *x* = 1) was synthesized at 500 °C for 1 h in air.

For the MnO_x electrodes, a Mn(NO₃)₂ (>97%) solution (2 M, dissolved in 10 μL ethanol) was evenly dispersed over the Ti foil using a short-haired brush. The electrode samples were dried at 60 °C for 10 min and annealed at various temperatures (200–500 °C) for 10 min in air.

This brushing-annealing cycle was repeated to load different masses of MnO_x ($1\text{--}10\text{ mg cm}^{-2}$). After the final cycle, the electrodes were additionally annealed for 1 h. Unless otherwise specified, MnO_x was annealed at $400\text{ }^\circ\text{C}$ with a loading mass of 7 mg cm^{-2} .

The surface morphologies of the as-synthesized samples (particles and/or electrodes) were analyzed by scanning electron microscopy (SEM, Su-8230, Hitachi) at an accelerating voltage of 5.0 kV . A high-resolution field-emission transmission electron microscope (HR-FE-TEM, JEOL, JEM-2100F) equipped with energy-dispersive spectroscopy (EDS, X-MaxN80, Horiba) was employed at an accelerating voltage of 200 kV . The surface areas of the powders were determined using the Brunauer–Emmett–Teller (BET; Micrometrics, ASAP 2020) method with N_2 as the adsorbate gas. X-ray diffraction (XRD, PANalytical EMPUREAN) with $\text{Cu-K}\alpha$ radiation (40 kV and 30 mA) and X-ray photoelectron spectroscopy (XPS, Thermo Fisher Scientific) with $\text{Al-K}\alpha$ ($h\nu = 1486.6\text{ eV}$) were performed to examine the crystalline patterns and elemental states of the as-synthesized samples, respectively.

Electrochemical activity tests and characterization

Linear sweep voltammograms of the as-synthesized electrodes (working electrodes) were obtained at a scan rate of 5 mV s^{-1} with a saturated calomel electrode (SCE, reference electrode) and Pt wire (counter electrode) using an electrochemical workstation (Ivium) in aqueous saline solutions with various compositions at pH 6 ($0.0137\text{--}0.941\text{ M NaCl}$, 0.5 M NaCl with 5 mM NaClO , and 0.4 M NaCl with 100 mM NaClO) or artificial seawater (Instant Ocean, salinity 36 g L^{-1}) at pH 8.2. NaClO_4 ($0.0137\text{--}0.5\text{ M}$) was used instead of NaCl to examine the chloride effect where necessary. For bulk electrolysis, the as-synthesized PRT or MnO_x electrodes were

held at constant J values (10–800 mA cm⁻²) in saline and seawater solutions in a single-compartment cell or two-compartment cell divided with a cation exchange membrane (CEM, CMI-7000S, Membranes International) (open or sealed to the atmosphere) while recording the electrode potentials (E) with respect to the SCE. When the PRT and MnO_x were coupled as the anode and cathode, respectively, a constant cell J (J_{cell}) of 80 mA cm⁻² was applied, and the resulting cell voltage (E_{cell}) was recorded using a direct current power supply (Agilent E3633A). During the bulk electrolysis, the solution was intermittently sampled and the free chlorines (represented by HClO/ClO⁻) were quantified using the *N,N*-diethyl-*p*-phenylenediamine reagent by the Hach method, as described elsewhere.^{15-17,32,33} The evolved gases in the cell headspace were also analyzed for O₂ and H₂ using a gas chromatograph (GC, Agilent 7820A; Inficon Micro GC Fusion) equipped with dual flame ionization and thermal conductivity detectors. For *in situ*, real-time analysis of O₂ and H₂ gases during prolonged bulk electrolysis over 100 h, the PRT anode/ MnO_x cathode pair at a distance of 30 mm was placed in an air-tight single-compartment cell through which N₂-sparged seawater (pH 8.2) continuously flowed at a rate of 11 mL min⁻¹ using a peristaltic pump (BT100-2J-19). **During electrolysis, evolved gases with high-purity N₂ carrier gas (>99.99%) flowed through the headspace of the cell at a rate of 20 mL min⁻¹ and entered into a quadrupole mass spectrometer (Quality Mass Spectrometer, Hiden, GB/QIC-20). The composition of the headspace O₂ and H₂ gases was calibrated using standard gases. Simultaneously, free chlorine in the electrolyzed seawater was analyzed.**

The electrochemical impedance spectroscopy of the as-synthesized PRT and MnO_x electrodes was conducted at 1.1 and -1.5 V vs. SCE in the frequency ranges of 100 kHz to

0.1 Hz and 50 kHz to 0.01 Hz, respectively, with an amplitude of 10 mV. The double-layer capacitance values of the electrodes proportional to the electrochemically active surface area were also examined by cyclic voltammetry in the E ranges of 0.85 to 0.95 V vs. SCE and -0.2 to -0.1 V vs. SCE, respectively, at varying scan rates in the range 20–100 mV s^{-1} . The Faradaic efficiencies (FEs) for the ClOR, ClRR, HER, and OER were estimated using the following equation:

$$\text{FE (\%)} = [\Delta P_c \times n_c \times F \times 100\%] / (J \times t)$$

where ΔP_c , n_c , F , and t refer to the amount (mol) of compounds produced (HClO , H_2 , and O_2) or consumed (HClO), stoichiometric number of such compounds ($n_c = 2$ for HClO and H_2 ; $n_c = 4$ for O_2), Faraday constant ($96,486 \text{ C mol}^{-1}$), and time (s), respectively. Hereafter, SCE is omitted for working electrodes with half-reactions, unless otherwise mentioned.

Results and discussion

Achieving high efficiency ClOR with durable PRT electrodes

The crystalline structures of the as-synthesized binary PR ($\text{Pt}_{0.2}\text{Ru}_{0.8}\text{O}_y$) and ternary PRT (PRT-0.3) particles were examined (**Fig. 1a**). The PR sample (annealed at $500 \text{ }^\circ\text{C}$) exhibited characteristic diffraction patterns of metallic Pt (PDF no. 04-0802 with $2\theta = 39.76^\circ$, 46.23° , 67.45° , 81.29° , and 85.71° for (111), (200), (220), (311), and (222), respectively) and RuO_2 (PDF no. 40-1290 with $2\theta = 28.01^\circ$, 35.05° , 40.02° , 40.06° , and 54.25° for (110), (101), (200), (111), and (211), respectively). Traces of metallic Ru peaks were also found at $2\theta = 42.15^\circ$ (002) and 44.01° (101) (PDF no. 06-0663), which disappeared upon oxidative annealing at a higher temperature (**Fig. S1**). The PRT sample annealed at $500 \text{ }^\circ\text{C}$ displayed a mixture of RuO_2

and rutile TiO₂ (PDF no. 21-1276 with 2θ = 27.45°, 36.09°, 41.23, 54.32°, 56.64°, and 69.01° for (110), (101), (111), (211), (220), and (301), respectively), along with Pt diffraction peaks.

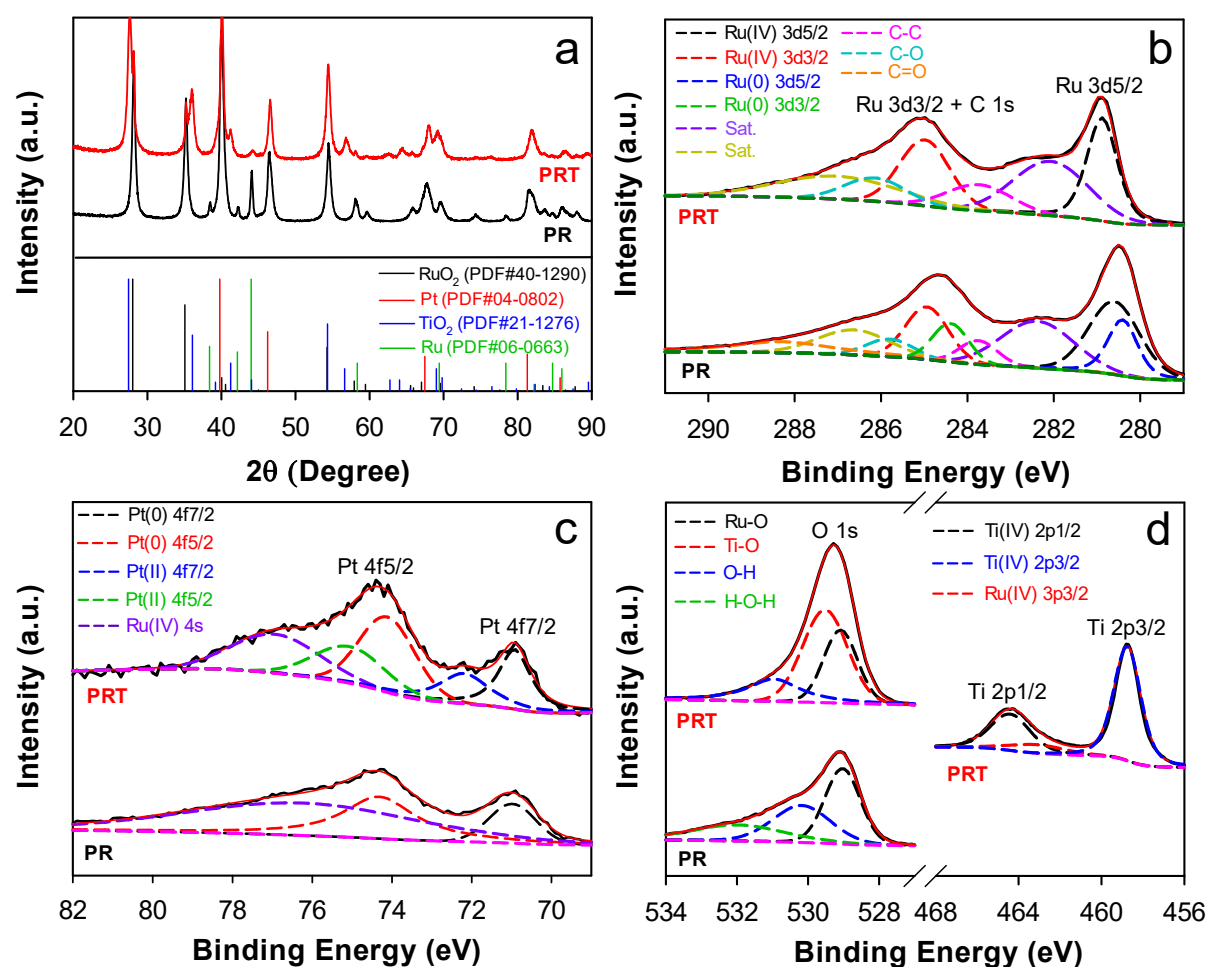


Figure 1. Surface characterization of as-synthesized binary PR (Pt_{0.2}Ru_{0.8}O_y) and ternary PRT (Pt_{0.06}Ru_{0.24}Ti_{0.7}O_y) particles annealed at 500 °C. (a) XRD spectra. (b-d) XPS spectra (b: Ru 3d, c: Pt 4f, and d: Ti 2p and O 1s). Deconvoluted XPS spectra are shown as dashed lines

The XPS spectra further revealed the elemental states of the as-synthesized PR and PRT particles (**Fig. 1b–1d**). For both samples, the Ru 3d spectra showed the spin-orbit doublet at binding energies of 280.9 eV (Ru 3d_{5/2}) and 285 eV (Ru 3d_{3/2}) (**Fig. 1b**). The Ru 3d_{3/2} band appeared to be mixed with the C 1s band, which was attributed to intrinsic or adsorbed carbon impurities. This was confirmed by the C 1s band observed for the PR sample annealed at 800 °C

(**Fig. S2**). Notably, the binding energy of the Ru 3d bands with the PRT sample was higher than that of the PR sample by 0.4 eV, owing to the dominant Ru⁴⁺ species and absence of Ru⁰ in the PRT sample. The Pt 4f bands in the PRT sample were located at 70.9 eV (Pt 4f_{7/2}) and 74.6 eV (Pt 4f_{5/2}), and the deconvolution of the bands revealed the coexistence of Pt⁰ and Pt²⁺ (**Fig. 1c**). In contrast, only Pt⁰ was found in the PR sample, and Pt²⁺ species were formed upon calcination at a high temperature (800 °C) (**Fig. S2**). The broad shoulder bands at ~76.4 eV were attributed to the Ru⁴⁺ 4s band. This suggests that TiO₂ facilitated the oxidation of Ru³⁺ and Pt⁰ to Ru⁴⁺ and Pt²⁺, respectively, during the oxidative annealing process at 500 °C. Moreover, the shifts in the Ru 3d and Pt 4f bands upon the introduction of TiO₂ indicate strong interactions between Ru and Pt with TiO₂. The Ti 2p band of the PRT sample also shows the characteristic doublet spectra of 2p_{3/2} (458.8 eV) and 2p_{1/2} (464.3 eV), attributed to the Ti⁴⁺ species (**Fig. 1d**). The contribution of the Ru 3p_{3/2} band was observed in the deconvoluted Ti 2p band. Finally, the O 1s bands of the PR and PRT samples were compared. The deconvoluted three sub-bands in the PR sample were assigned to lattice oxygen (Ru-O-Ru at 529.1 eV), surface oxygen (Ru-OH at 530.2 eV), and adsorbed oxygen (e.g., H₂O at 532.1 eV).³⁴ Although an additional lattice oxygen (Ti-O-Ti at 529.7 eV) was found in the PRT sample, the contribution of the adsorbed oxygen was negligible.

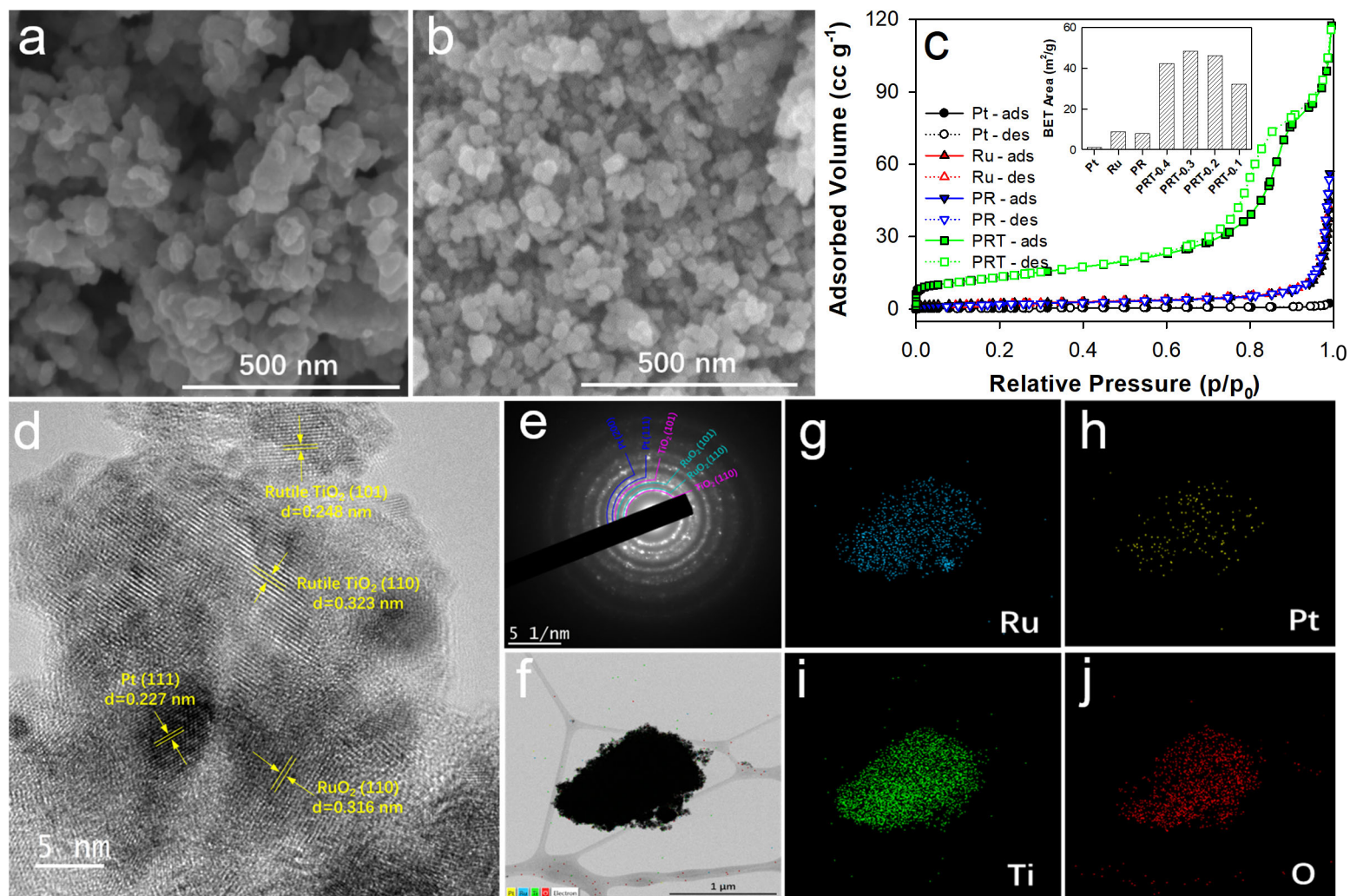


Figure 2. (a and b) SEM images of as-synthesized PR and PRT particles, respectively. (c) Isotherm N₂ adsorption-desorption curves with the as-synthesized Pt, RuO₂, PR, and PRT samples. Inset shows BET surface areas. (d and e) HR-TEM image and SAED pattern of the PRT sample. (f–j) EDS and elemental mappings of the PRT sample.

The as-synthesized PR sample was composed of irregularly shaped particles (**Fig. 2a**), and the primary particle size increased with the annealing temperature (**Fig. S3**). In contrast, the PRT sample displayed a relatively uniform spherical configuration with an average diameter of ~ 27 nm (**Fig. 2b**). The isotherm N_2 adsorption-desorption curves indicated a negligible interaction of N_2 with Pt, whereas N_2 adsorption was enhanced with Ru (and consequently PR) (**Fig. 2c**). Notably, the PRT sample exhibited a significantly high adsorption capacity with hysteresis (**Fig. S4**), indicating a mesoporous structure. The BET areas with PRT were large (>40 $m^2 g^{-1}$), whereas the other samples exhibited areas of <10 $m^2 g^{-1}$ (**Fig. 2c inset**). The HR-TEM analysis of the PRT sample showed aggregates with primary particles of 10–20 nm (**Fig. S5**). Magnified HR-TEM images further revealed mixed interplanar spacings of 0.248 and 0.323 nm corresponding to the (101) and (110) planes of rutile TiO_2 , respectively, 0.316 nm corresponding to the (110) plane of RuO_2 , and 0.227 nm corresponding to the (111) plane of Pt (**Fig. 2d**). These crystalline planes were intimately connected to each other, forming strong interfacial contacts. The selected area electron diffraction (SAED) pattern further confirmed the coexistence of ternary crystalline structures with TiO_2 , RuO_2 , and Pt (**Fig. 2e**), which is in good agreement with the XRD and HR-TEM images. The EDS and elemental mapping analysis clearly showed a uniform distribution of Ru, Pt, Ti, and O; however, the Pt signal was weak owing to the low Pt content ($\sim 6\%$) (**Fig. 2f–2j**).

The as-synthesized binary PR and ternary PRT particles were sprayed onto Ti meshes, and the electrochemical behavior of the as-sprayed electrodes was examined in NaCl and $NaClO_4$ electrolytes. The linear sweep voltammograms with the PR electrode showed an onset potential (E_{on}) value of ~ 0.95 V in both electrolytes, whereas a larger J was obtained in the NaCl

electrolyte (**Fig. 3a**). A similar tendency was also observed with single Ru (i.e., RuO₂) and Pt electrodes; however, the Pt electrode was nearly inactive in both solutions. The ternary PRT followed the same tendency, but the voltammogram difference between NaCl and NaClO₄ was significant. In the NaCl solution, the optimal PR fraction was found to be 0.3 (i.e., PRT-0.3 or Pt_{0.06}Ru_{0.24}Ti_{0.7}O_y), beyond which the electrochemical activity decreased (**Fig. S6**). Considering that OER is the primary oxidation reaction in a chloride-free solution (i.e., NaClO₄), whereas a combination of ClOR and OER can occur in a chloride solution, an enhanced *J* value with NaCl may indicate that ClOR was favored over OER with PR and more significantly with PRT.

To quantitatively compare the ClOR efficiency, bulk electrolysis with various electrodes at *J* = 80 mA cm⁻² was performed (**Fig. 3b**). In 0.5 M NaCl, free chlorine (i.e., HClO/ClO⁻) was linearly produced with time (**Fig. 3b inset**), maintaining a ClOR-FE value of ~100% with the PRT electrode. This FE value was higher than that with the PR electrode (~84%), although the molar amounts of Pt and Ru in the PRT electrode were only 30% of those in the PR electrode. Moreover, Pt and Ru exhibited considerably lower ClOR-FE values (<60%). As the electrolyte concentration decreased to 0.137 and 0.0137 M, respectively, the voltammogram difference with PRT between NaCl and NaClO₄ gradually reduced (**Fig. S7**). Nevertheless, ~100% ClOR-FE with PRT was obtained in 0.137 M NaCl. When the loading amount of PRT (3.33 mg cm⁻²) was reduced by 75% (0.83 mg cm⁻²; denoted as PRT (1/4)), *E*_{on} increased by +0.08 V and *E* increased to 0.5 V for *J* = 80 mA cm⁻². This larger *E* value was attributed to the reduced active surface area of the catalysts.

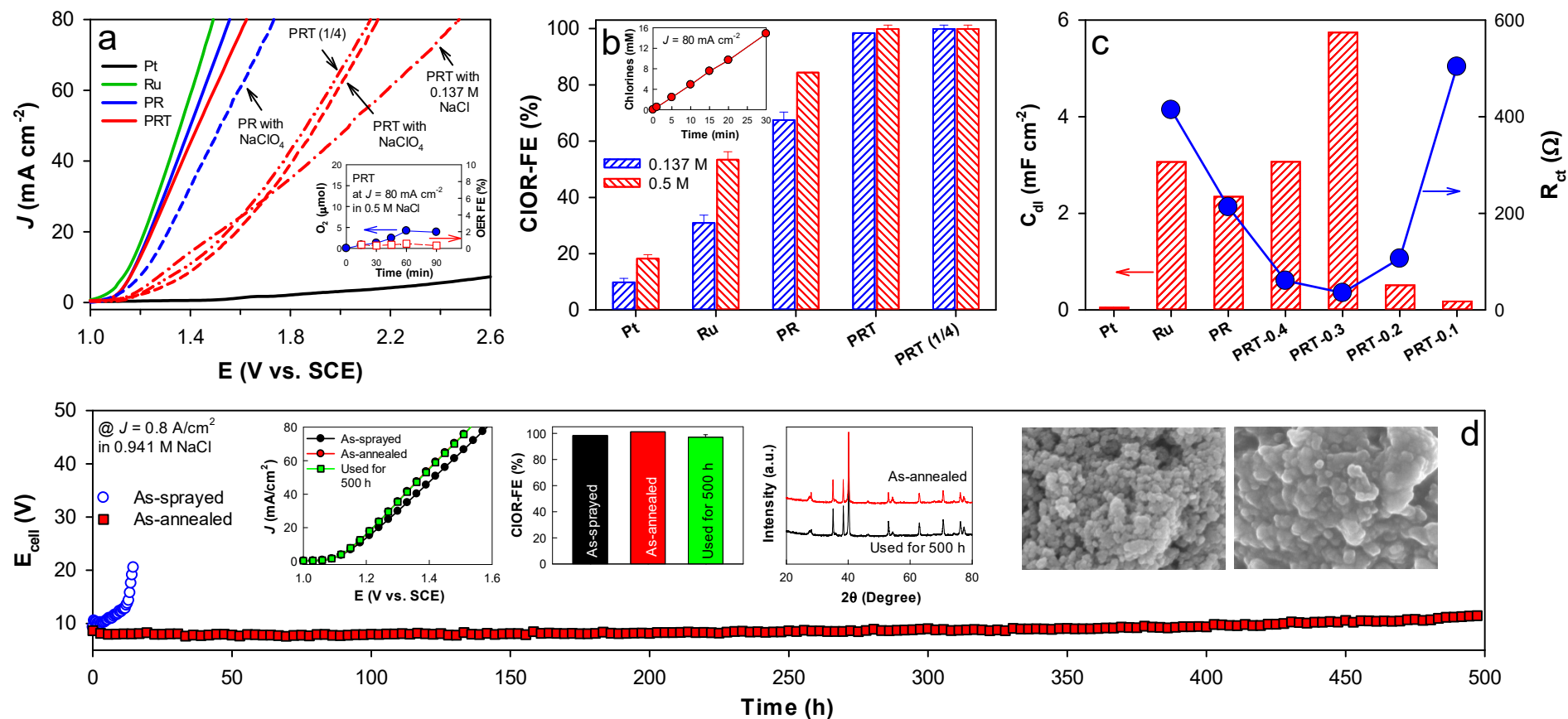


Figure 3. (a) Linear sweep voltammograms with Pt, Ru, PR, and PRT electrodes in 0.5 M NaCl solutions at pH 6. For comparison, 0.137 M NaCl or 0.5 M NaClO₄ (pH 6) were also used. PRT (1/4) refers to the PRT with loading mass of 0.83 mg cm⁻². The inset shows the O₂ evolution and its FE with PRT at $J = 80 \text{ mA cm}^{-2}$ in 0.5 M NaCl. (b) FE values of ClOR at $J = 80 \text{ mA cm}^{-2}$. The inset shows the free chlorine production with time at $J = 80 \text{ mA cm}^{-2}$. (c) C_{dl} and R_{ct} values. (d) Prolonged electrolysis with the as-sprayed and as-annealed PRT anodes at $J = 800 \text{ mA cm}^{-2}$ in 0.941 M NaCl. Insets show voltammograms and FE values (at $J = 80 \text{ mA cm}^{-2}$) of the as-sprayed, as-annealed, and 500 h-used PRT electrodes in 0.5 M NaCl solutions. XRD spectra and SEM images (left: as-annealed, right: used for 500 h) are also shown.

For example, the masses of the Pt and Ru salts with PRT (1/4) decreased to 0.073 and 0.135 mg cm⁻², respectively. Nevertheless, ~100% ClOR-FE was achieved with PRT (1/4) (**Fig. 3b**). Under the same electrolysis conditions, the OER with PRT was remarkably inhibited with an FE of <1% (**Fig. 3a inset**). These results indicate that the as-synthesized PRT was highly selective for ClOR.

The electrocatalytic activity of the as-synthesized electrodes is associated with the electrochemical active surface area, which is proportional to the double-layer capacitance (C_{dl}). The C_{dl} value for PRT-0.3 was estimated to be ~5.7 mF cm⁻², which was higher than the values for single and binary composition electrodes (0.05, 3.1, and 2.3 mF cm⁻² for Pt, Ru, and PR, respectively) and also the values for other ternary PRT electrodes (**Figs. 3c and S8**). C_{dl} appeared to be influenced by oxides (TiO₂ and RuO₂) and was maximized at $x = 0.3$. PRT-0.3 also exhibited the lowest interfacial charge-transfer resistance (R_{ct}) (**Fig. S9**). Electrochemical impedance spectroscopic analysis revealed that PRT-3 had the smallest R_{ct} value (35.7 Ω) among the PRT series. The single and binary electrodes exhibited large R_{ct} values (Pt: 2700 Ω , Ru: 414 Ω , PR: 213 Ω). These results partly explain the superior activity of PRT-0.3 compared to that of all other single, binary, and ternary compositions.

The stability of the optimized PRT electrode was examined under severe conditions ($J = 800$ mA cm⁻²; 0.941 M NaCl) while recording E_{cell} of the PRT anode and the mesh-type Ti cathode pair (**Fig. 3d**). E_{cell} was 10.5 V in the initial stage and abruptly increased to >20 V at ~10 h. The PRT particles immobilized on the substrate (Ti mesh) via a spray process readily peeled off from the substrate after 10 h. Hence, the as-sprayed PRT electrode was annealed at 300 °C for 6 h to induce robust contact, and its stability was tested again. The initial value of

E_{cell} (8.5 V) was maintained over 400 h, and only slightly increased to ~11 V after 500 h. The voltammograms of the annealed PRT electrodes (before and after 500 h of electrolysis) were the same, whereas their J values were larger than those of the as-sprayed PRT, which was attributed to the enhanced contact of the catalysts with the substrate upon annealing (**Fig. 3d inset**). The ClOR-FE value of the 500 h-used PRT electrode was ~97%, which was nearly the same as that of the as-annealed electrode. In addition, the crystalline structure (XRD in **Fig. 3d inset**) and elemental states (XPS in **Fig. S10**) did not change after 500 h of electrolysis. However, the catalyst surface appeared to melt, and the particles became aggregated (SEM in **Fig. 3d inset**). This phenomenon of interparticle fusion is typical.³⁴⁻³⁶ Nevertheless, the as-observed electrocatalytic activity and stability were outstanding compared to those obtained in previous reports (**Table S1**).

Inhibiting ClRR with MnO_x electrodes

For suppression of the ClRR, MnO_x electrodes were synthesized at various temperatures. The MnO_x electrode synthesized at 200 °C (MnO_x-200) demonstrated an inhomogeneous cracked morphology, whereas compact uniform particles were observed in the MnO_x-400 sample (**Fig. S11**). The XRD spectra of both samples exhibited the same diffraction patterns corresponding to Mn₂O₃ (PDF no. 24-0508; $2\theta = 23.12^\circ, 32.92^\circ, 38.20^\circ, 45.14^\circ, 49.32^\circ, 55.14^\circ,$ and 65.72° for the (211), (222), (400), (323), (413), (044), and (622) planes, respectively) (**Fig. 4a** and **S12**). The HR-TEM images of the MnO_x-400 sample demonstrated an interplanar spacing of 0.2714 nm, corresponding to the (222) plane of Mn₂O₃ (**Fig. 4b**). XPS analysis was performed to examine the Mn valence states of both samples (**Fig. 4c**).

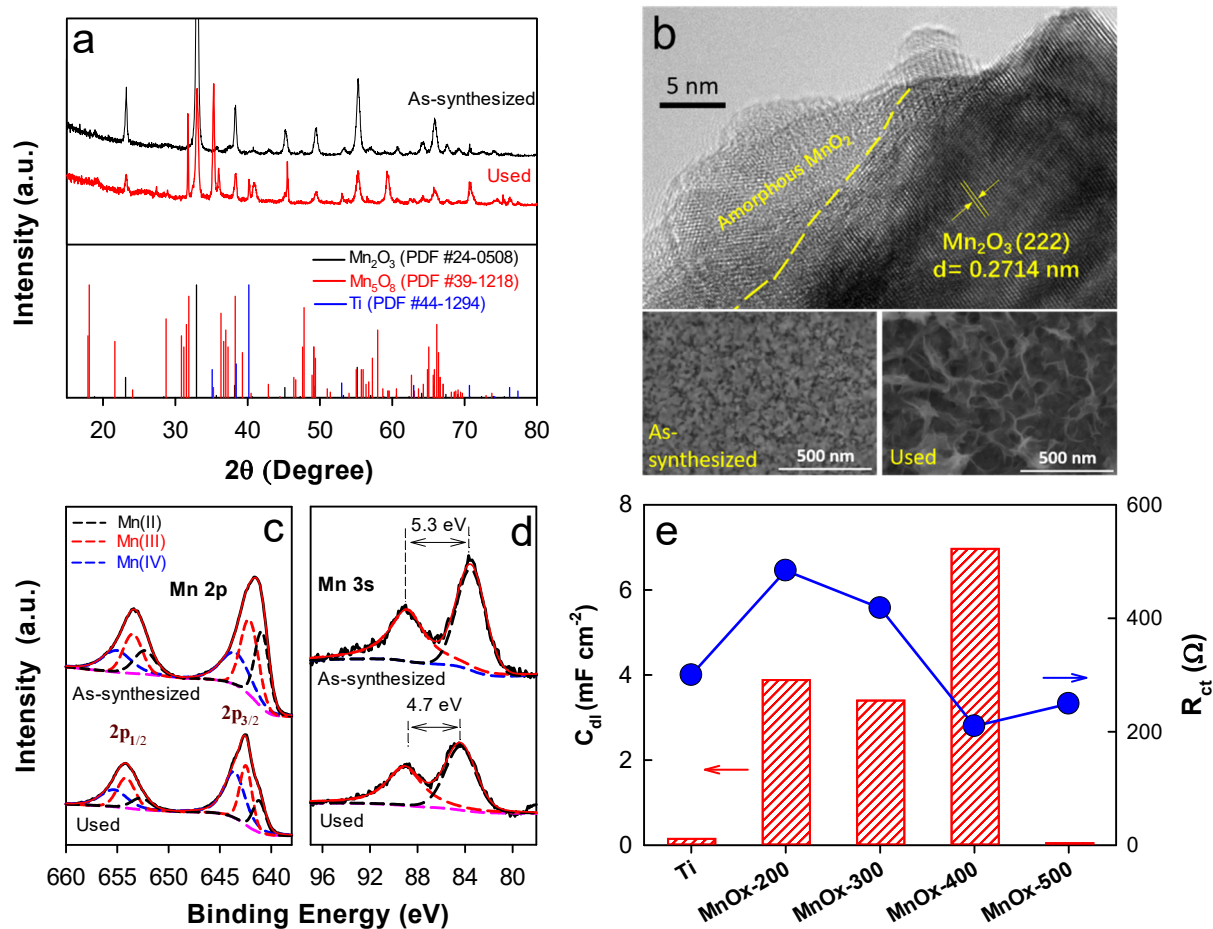


Figure 4. Surface and electrochemical analysis of MnO_x electrodes (deposited on Ti plates) annealed at various temperatures (200–500 °C). Unless otherwise specified, MnO_x stands for MnO_x-400 (annealed at 400 °C). (a) XRD spectra, (b) HR-TEM and SEM images (top and bottom panels, respectively), (c and d) XPS spectra (Mn 2p and Mn 3s, respectively), and (e) C_{dl} and R_{ct}. After 100 h of electrolysis, the used MnO_x samples were also examined with XRD, SEM, and XPS.

The Mn 2p bands (2p_{1/2} and 2p_{3/2} at binding energies of 653.4 eV and 641.6 eV, respectively) in both samples appeared similar, and the deconvolution of the bands with MnO_x-400 revealed mixed Mn(II), Mn(III), and Mn(IV) sub-bands at fractions of 30%, 37%, and 33%, respectively (31%, 39%, and 30% for MnO_x-200, respectively). The binding energy differences between the two Mn 3s bands were estimated to be 5.2 eV and 5.3 eV, respectively (Fig. 4d). This indicated that the average oxidation states of the surface Mn atoms were ~3.1 and ~3.0 for

MnO_x-200 and MnO_x-400, respectively.^{37,38} The as-synthesized MnO_x electrodes exhibited various C_{dl} and R_{ct} values depending on the annealing temperature (**Fig. 4e**). Among the samples, MnO_x-400 showed the highest C_{dl} value (6.97 mF cm⁻²) and the lowest R_{ct} value (209 Ω) (**Fig. S13** and **S14**). MnO_x-200 demonstrated the second highest C_{dl} (3.89 mF cm⁻²) but the worst R_{ct}. The C_{dl} of MnO_x-500 was as small as that of the Ti substrate, but the R_{ct} value was comparable to that of MnO_x-400.

The electrocatalytic activities of the as-synthesized MnO_x electrodes (deposited on a Ti plate) were examined for the CIRR and HER in NaCl solutions (with and without NaClO). The voltammograms of MnO_x-400 indicated an *E*_{on} of *approximately* -1.8 V (corresponding to -1.2 V vs. reversible hydrogen electrode (RHE)), regardless of the presence of NaClO (**Fig. 5a**). This indicates a negligible CIRR with HClO/ClO⁻. In contrast, the Ti plate (without MnO_x catalysts) exhibited an *E*_{on} of *approximately* -1.7 V (corresponding to -1.1 V vs. RHE) in a 0.5 M NaCl solution. The voltammogram gradually shifted to a positive potential in the presence of 5 mM and 100 mM NaClO, with *E*_{on} values of -1.6 and -1.4 V, respectively. Notably, two distinct reduction peaks were observed at -0.55 V and -1.04 V with 100 mM NaClO due to the CIRR.^{39,40} For comparison, voltammograms with Pt electrodes were also examined. *E*_{on} with a Pt electrode exhibited a more positive value than the corresponding values with Ti and MnO_x in the NaCl solution. In addition, the voltammogram shifted to a positive potential in the presence of NaClO, and CIRR peaks were observed. These behaviors indicate that Pt should also be active in the CIRR. Bulk electrolysis for the CIRR was further performed using 5 mM NaClO (**Fig. 5b**). For the Pt and Ti substrates at *J* = -10 mA cm⁻², the concentrations of chlorine (C_t / C₀) linearly decreased with time with a CIRR-FE of 18–20%.

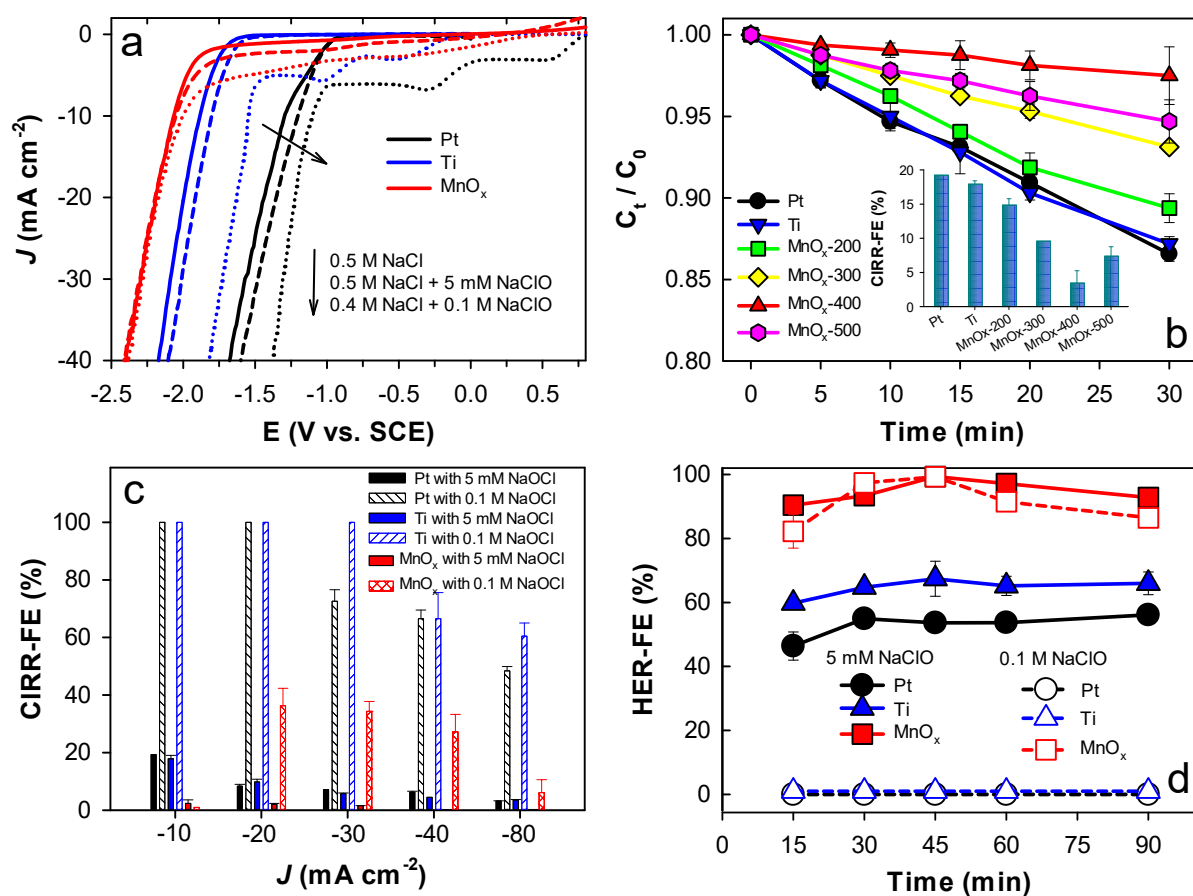


Figure 5. Electrochemical behavior and activities for CIRR and HER with Pt, Ti, and MnO_x electrodes in 0.5 M NaCl, 0.5 M NaCl with 5 mM NaClO, or 0.4 M NaCl with 0.1 M NaClO solutions at pH 6. Unless otherwise specified, MnO_x stands for MnO_x-400 (annealed at 400 °C). (a) Linear sweep voltammograms. (b) Changes in the concentration of chlorines (C_t / C_0) with various electrodes at $J = -10 \text{ mA cm}^{-2}$ in 0.5 M NaCl with 5 mM NaClO in a two-compartment cell divided by a CEM. The inset shows CIRR-FE values in 30 min. (c) CIRR-FE values as a function of J and NaOCl concentration. (d) HER-FE changes at $J = -10 \text{ mA cm}^{-2}$ in 0.5 M NaCl with 5 mM NaClO.

This decrease in chlorine concentration was significantly inhibited by the MnO_x electrodes.

Among them, MnO_x-400 required the smallest E value at $J = -10 \text{ mA cm}^{-2}$ (Fig. S15), with a CIRR-FE of approximately 3% (Fig. 5b inset). The CIRR typically proceeds with HClO/ClO⁻ adsorption, followed by breaking of the Cl–O bond, Cl adsorption with concurrent H₂O release, and Cl⁻ release.⁴¹ Considering that the isoelectric point of MnO_x was located at $\text{pH} < \sim 4$ (Fig. S16),⁴² the HClO/ClO⁻ access should be inhibited to the negatively charged MnO_x surface at a

circumneutral pH value.²⁴

The CIRR-FE was significantly influenced by the type of electrode, NaClO concentration, and J value (**Fig. 5c**). For the Ti substrate with 5 mM NaClO, the CIRR-FE value gradually decreased with increasing J because of the concurrent HER. With 100 mM NaClO, the CIRR-FEs with the Ti were $\sim 100\%$ at $J \leq -30 \text{ mA cm}^{-2}$ and decreased to $\sim 65\%$ at larger J values. A similar behavior was observed for Pt, albeit with lower CIRR-FE values than those for Ti at $J \geq -30 \text{ mA cm}^{-2}$ with 100 mM NaClO. This indicates that the HER occurred more efficiently with Pt than with Ti, particularly at high concentrations of NaClO. In contrast, the CIRR-FEs with MnO_x were $< 3\%$ at all the tested J values with 5 mM NaClO. With 100 mM NaClO, a volcanic CIRR-FE profile was obtained by blocking the HClO access at a low J and by a competitive HER at a high J . MnO_x , Ti, and Pt electrodes produced H_2 in NaCl solutions with 5 mM NaClO at $J = -10 \text{ mA cm}^{-2}$, leading to HER-FEs of $> 90\%$, $\sim 65\%$, and $\sim 55\%$, respectively (**Fig. 5d**). When the NaClO concentration increased to 100 mM, no H_2 was evolved with Ti and Pt (hence, zero HER-FEs). However, nearly the same amounts of H_2 were produced with MnO_x with 5 mM and 100 mM NaClO (hence, similar HER-FEs). These results confirm that MnO_x is highly selective for the HER in saline water containing chlorines.

The stability of the as-synthesized MnO_x electrode was further tested over 100 h at $J = -10 \text{ mA cm}^{-2}$ and -80 mA cm^{-2} in 0.5 M NaCl solutions. E values of approximately -3.5 V and -4.2 V , respectively, were maintained over 100 h (**Fig. S17**). The used MnO_x electrode with $J = -10 \text{ mA cm}^{-2}$ for 100 h exhibited new crystalline structures originating from Mn_5O_8 (**Fig. 4a**). The particle-compact surface also changed into an unfolded tissue-like surface (**Fig. 4b**). A deconvolution of the Mn 2p bands further revealed a fractional ratio of 17/32/51 for

Mn(II)/Mn(III)/Mn(IV) (**Fig. 4c**), whereas the average oxidation state of the Mn atom increased from 3.0 to ~3.7 (**Fig. 4d**). Despite the cathodic reaction, this increase in the oxidation state can be attributed to the hypochlorite-mediated partial oxidation of MnO_x.²⁴

Membraneless seawater electrolysis and production of pure H₂ gas

The as-synthesized PRT and MnO_x electrodes were wired in a membraneless single-compartment cell containing seawater at a pH of 8.2. The E_{on} values in the anodic and cathodic voltammograms were slightly different from those in 0.5 M NaCl at pH 6 (**Fig. 6a**), likely owing to different pH values and composition.⁴³ The voltammograms further indicated that an anode/cathode cell potential (E_{cell}) of ~4.8 V (= 1.8 V – (–3 V)) was required for $J_{\text{cell}} = 80 \text{ mA cm}^{-2}$. Under galvanostatic conditions at $J_{\text{cell}} = 80 \text{ mA cm}^{-2}$, chlorine was linearly produced with a ClOR-FE of ~100% (**Fig. 6b**). This FE value was the same as that obtained for the two-compartment cell equipped with a membrane (**Fig. 3**). When MnO_x was replaced with a Ti plate, the ClER-FE gradually decreased to ~80% owing to the concurrent ClRR with the Ti plate. This ClRR should compete with the HER, thereby lowering the HER-FE to ~80% (**Fig. 6c**). In contrast, an HER-FE of ~100% was obtained with the PRT/MnO_x pair.

Prolonged electrolysis of seawater was conducted in a continuous-flow, membraneless reactor equipped with a PRT/MnO_x pair (**Fig. 6d**). The headspace gases were monitored *in situ* in real time using a quadrupole mass spectrometer. At $J_{\text{cell}} = 80 \text{ mA cm}^{-2}$, an E_{cell} of ~4.8 V was stably maintained over 100 h, whereas ClOR occurred at an FE of ~100%, and the OER-FE was <0.5% (virtually zero). H₂ gas was continuously produced with an FE of approximately 100%. No changes in the E_{cell} and FE values were observed during electrolysis. These results

clearly indicate that the PRT/MnO_x pair is effective in producing high-purity H₂ gas via membraneless seawater electrolysis.

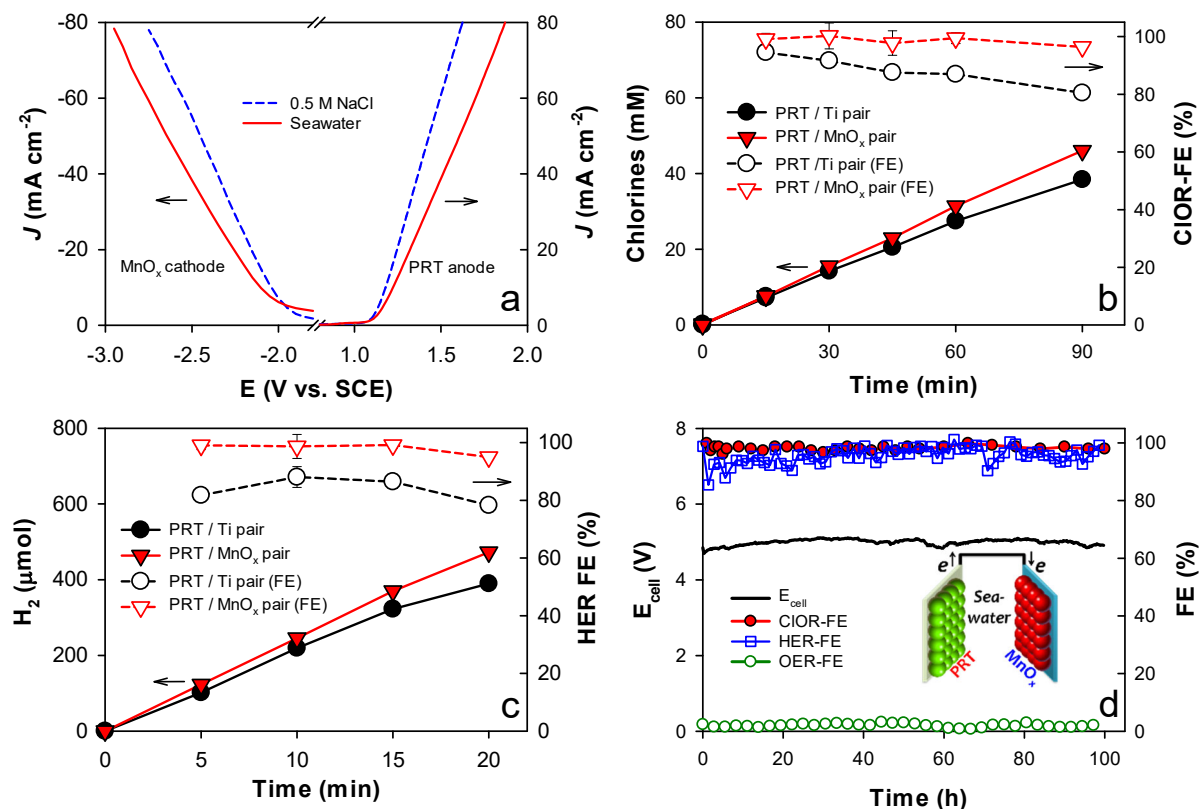


Figure 6. Seawater electrolysis with a PRT anode and MnO_x cathode (PRT/MnO_x) pair in a single-compartment cell. (a) Linear sweep voltammograms of each electrode in 0.5 M NaCl and seawater solution (salinity 36 g L⁻¹). (b) Production of chlorines and ClOR-FEs with PRT/Ti and PRT/MnO_x pairs at $J_{\text{cell}} = 80$ mA cm⁻² in seawater solution. (c) Production of H₂ and HER-FEs with PRT/Ti and PRT/MnO_x pairs at $J_{\text{cell}} = 80$ mA cm⁻² in seawater solution. (d) Prolonged electrolysis with the PRT/MnO_x pair at $J_{\text{cell}} = 80$ mA cm⁻² in seawater solution. Evolved H₂ and O₂ gases were *in situ* real time measured while chlorines were intermittently analyzed. Changes in E_{cell} were also recorded.

Conclusions

This study successfully demonstrated the production of high-purity H₂ using a ClOR-selective anode and HER-selective cathode pair via membraneless electrolysis of undisturbed, unbuffered seawater. An optimized ternary PRT electrocatalyst with a minimized Pt level exhibited superior activity compared with single and binary component catalysts for the selective production of chlorines in saline and seawater, with complete inhibition of the OER. Such an activity can be attributed to the excellent surface and electrochemical characteristics of the PRT electrocatalyst, including a large surface area and low interfacial charge-transfer resistance. The PRT electrocatalyst maintained industry-level stability with constant ClOR activity over 500 h under severe conditions. Moreover, MnO_x drove the HER at an FE of ~100% in the presence of free chlorines while completely inhibiting the ClRR. In contrast, the ClRR and HER occurred concurrently with other commonly used electrodes (Ti and Pt), reducing the HER efficiency. The observed selectivity of the HER with MnO_x was associated with the limited interaction of free chlorines. Finally, the PRT anode/MnO_x cathode pair produced H₂ gas at an FE of ~100% via seawater electrolysis at pH 8.2 over 100 h in a membraneless reactor. No O₂ was evolved, whereas ClOR occurred at an FE of ~100% during seawater electrolysis. The as-developed electrolyzer does not require the use of membranes for producing high-purity H₂ with an FE of ~100% via unbuffered natural seawater electrolysis.

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